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Fingerprinting mean composition of lithium polysulfide standard solutions by applying high energy resolution fluorescence detected X-ray Absorption Spectroscopy

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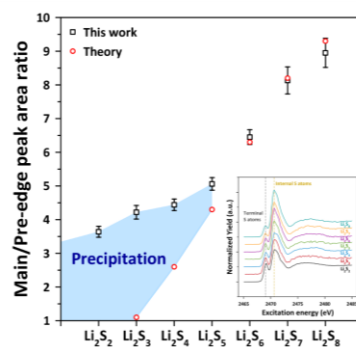
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ABSTRACT

In a lithium/sulfur (Li/S) battery, the reduction of sulfur during discharge involves a particular mechanism, where the active material successively dissolves into the electrolyte to form lithium polysulfide intermediate species (Li_2S_x), with x being a function of the state of charge. In this work, sulfur K-edge Resonant Inelastic X-ray Scattering measurements were performed for the characterization of different Li_2S_x polysulfide standard solutions. High Energy Resolution Fluorescence Detected X-ray Absorption Spectroscopy allowed clear separation the pre-edge absorption peak corresponding to terminal sulfur atoms from the main absorption peak due to internal atoms, and to evaluate quantitatively the evolution of the peak area ratio as a function of the polysulfide chain length. Results of this experimental work demonstrate that the normalized area of the pre-edge is a reliable fingerprint of Li_2S_x mean chain length in agreement with recent theoretical predictions. As a perspective, this work confirms that *operando* HERFD XAS can be used to differentiate mean polysulfide composition, which is key issue in the characterization of Li/S cells.

TOC GRAPHICS



KEYWORDS

Lithium polysulfide standards; Sulfur K-edge RIXS measurements, sulfur HERFD-XAS spectra, Polysulfide mean chain length

Lithium/sulfur (Li/S) batteries are one of the most attractive rechargeable energy storage systems, thanks to the low cost and abundance of elemental sulfur, as well as the high theoretical gravimetric energy density attainable.¹ This post lithium-ion system usually relies on the use of a lithium metal electrode, an organic electrolyte and a sulfur based cathode where the elemental sulfur (S₈) is reduced to produce solid Li₂S. The electrochemical activity of sulfur (or of the lithiated Li₂S counterpart) involves a particular cycling mechanism, where the active material successively dissolves into the electrolyte with the formation of lithium polysulfide (PS) intermediate species (Li₂S_x, 2 ≤ x ≤ 8), with x being a function of the state-of-charge, and precipitates at the positive electrode (S₈ and Li₂S). Thus, the electrochemical processes of Li/S batteries are complex, and still subject to debate.

Different discharge mechanisms have already been proposed, involving different lithium PS intermediate species and reaction pathways.^{2,3,4,5} However, the complex chemical and electrochemical equilibria, as well as the lack of analytical techniques to unambiguously determine the exact composition of Li₂S_x, make the determination of a univocal discharge mechanism very difficult. High-performance liquid chromatography has been applied *ex situ*, and offers a good selectivity towards each polysulfide.^{2,3} However, the detailed understanding of the discharge mechanism requires the use of *operando* methods, with both good chain length selectivity and time resolution. To achieve this, *operando* UV-vis and X-ray Absorption (XAS) spectroscopy techniques have been applied to Li/S cells, providing partial information on Li₂S_x intermediate species.^{3,4,5,6} In particular, *operando* X-Ray Absorption Near Edge Structure (XANES) spectroscopy, which is capable of detecting both solid and liquid phases, has been successfully applied in several Li/S battery studies,^{5,7,8} giving information relative to the oxidation state of sulfur species including Li₂S_x during the battery cycle.

A linear combination fitting analysis was used in these studies to extract relative amounts of the three main sulfur compounds formed in the battery. Reference spectra used in the fit were

either obtained by measuring Li_2S_x standards or Na_2S_x substitutes⁵, or in some cases extracted directly from a set of operando spectra⁷. While the total relative amount of polysulfides produced within the battery could be followed during the cycle, it was only possible to make a rough qualitative estimate of the polysulfide chain length. However, a recent theoretical study¹⁵ suggested a linear relationship between the peak areas of the two main spectral features and the polysulfide chain length, which could be used to differentiate the polysulfides from XANES spectra. In such measurements, the energy resolution of the measured absorption spectra is ultimately limited by the core-hole lifetime broadening and this represents an obstacle in resolving clearly the main features in the Li_2S_x absorption spectrum. Compared to standard fluorescent mode XAS measurements, energy resolution can be enhanced below the core-hole lifetime broadening using Resonant Inelastic X-ray Scattering (RIXS). In this case, the fluorescence is detected by the emission spectrometer with energy resolution comparable to the resolution of the beamline monochromator. The overall spectral broadening is in this case limited mainly by the experimental resolution (the final state broadening is usually almost negligible), and a significant sharpening effect can be observed in the RIXS spectra. While RIXS in the hard X-ray range is commonly exploited to study 3d transition complexes⁹, the tender X-ray range containing the sulfur absorption edge has been much less explored, mainly due to several technical challenges inherent to measurements in this low energy range.

In our previous work,¹⁰ *operando* RIXS measurements at the sulfur K-edge were successfully applied to enhance the sensitivity for lithium polysulfide detection, by exciting the polysulfides resonantly compared to the solid S_8 and Li_2S counterparts. In the present work, sulfur K-edge RIXS measurements were used for the characterization of different Li_2S_x polysulfides. In order to closely mimic the environment in a Li/S cell, measurements were performed on dissolved lithium PS standard solutions. Enhanced spectral resolution was used to decompose the main

spectral features and determine precisely their area ratios, which can be directly compared to theoretical predictions.¹⁵

An example of a full K-L RIXS map is shown in Fig. 1(a) for the Li_2S_2 equivalent standard solution. The map was built by recording a series of 100 consecutive sulfur $\text{K}\alpha$ emission spectra while changing the excitation energy across the sulfur K absorption edge in 0.2 eV steps. Due to the high experimental energy resolution, a significant narrowing can be observed in the measured RIXS map, with the first pre-edge resonance due to terminal sulfur atoms well resolved from the main absorption peak corresponding to internal sulfur atoms. By making a horizontal cut through the measured RIXS map at fixed emission energy corresponding to the top of the $\text{K}\alpha_1$ emission line, an absorption like spectrum with superior resolution is obtained. This technique is referred as High Energy Resolution Fluorescence Detected XAS (HERFD-XAS),^{11,12} and can be used to overcome the intrinsic core-hole lifetime broadening limitation of the XAS spectroscopy. This is demonstrated in Fig. 1(b), comparing the measured Li_2S_2 sulfur K-edge HERFD-XAS spectrum with the standard XAS spectrum measured as well. Significant improvement in spectral resolution is observed in Fig. 1(b).

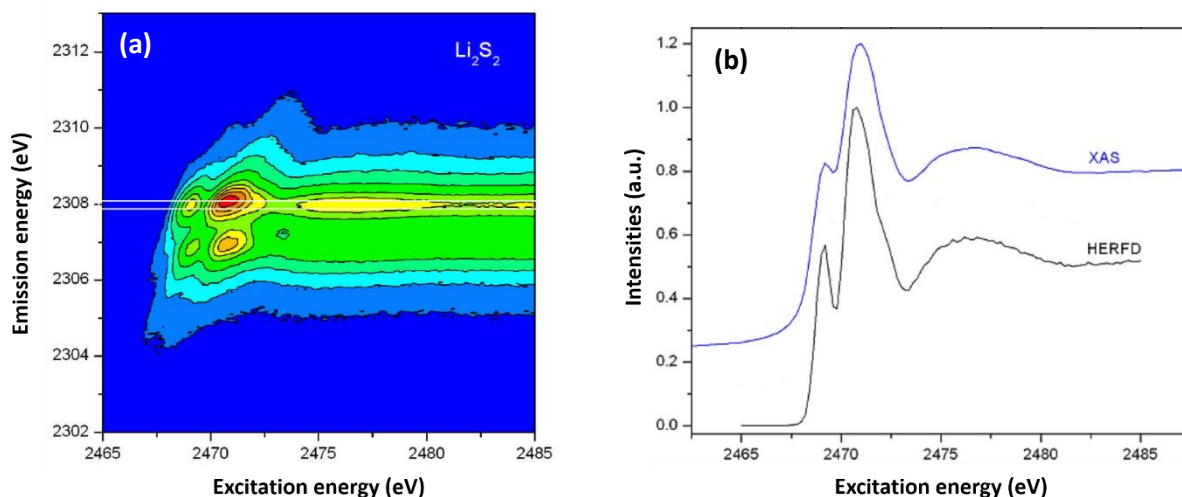


Figure 1. RIXS map of Li_2S_2 $10^{-1} \text{ mol.L}^{-1}$ standard solution in TEGDME/DIOX 1/1. The vertical axis corresponds to the emission energy around K-L (K_α) transition and horizontal axis corresponds to the excitation energy around the K absorption edge of sulfur (a). Comparison of experimental spectra obtained by standard XAS (in blue) and by horizontal cut through the RIXS map (in black) (b).

Compared to standard XAS, the increased resolution of HERFD XAS spectra allows the pre-edge peak attributed to terminal atoms of lithium PS ($\sim 2469 \text{ eV}$) to be clearly resolved from the main peak ($\sim 2471 \text{ eV}$) generally attributed to elemental sulfur (S-S bond in S_8 ring).^{10,18,13,14,15} Observation of Fig. 2 shows as predicted that the relative area (resp. intensity) of the pre-edge peak increases while the polysulfide chain length decreases. Quantitative analysis can be performed by determining the area (resp. intensity) of the pre-edge peak relative to that of the main one, to better differentiate the polysulfide species.

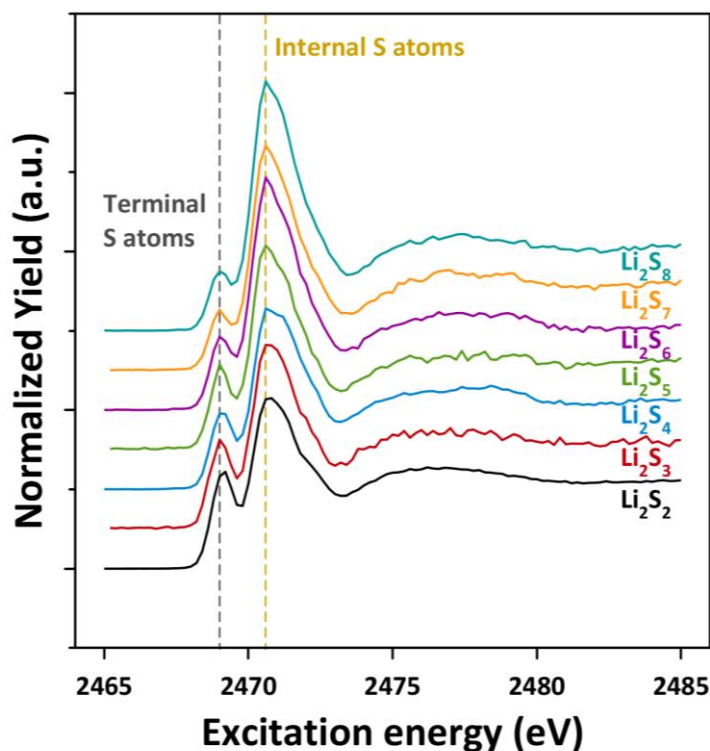


Figure 2. Normalized HERFD sulfur K-edge absorption spectra recorded for different lithium PS standard solutions Li_2S_x ($x = 2 \dots 8$) at $10^{-1} \text{ mol.L}^{-1}$ in TEGDME/DIOX 1/1. The spectra are shifted vertically for clarity.

In a recent comprehensive theoretical work of Pascal *et al.*,¹⁵ sulfur XAS spectra of lithium PS dissolved in TEGDME were simulated using first-principles molecular dynamics calculations. Their calculations confirm that the two main absorption features arise from differently coordinated sulfur atoms in the polysulfide molecule. The pre-edge feature is attributed to terminal sulfur atoms associated to lithium atoms at either end of the polysulfide and the main-edge to the internal sulfur atoms. Density functional theory (DFT) calculations revealed increased local electron density (charge) of the terminal atoms compared to the internal ones, leading to reduced binding energy of the core electron and consequently the shift of the corresponding absorption feature to lower energies with respect to those corresponding to internal atoms in the polysulfide chain. This attribution of main absorption features was

confirmed also by the work of Kosugi *et al.*,¹⁶ who have carried out *ab initio* calculations on XAS spectrum of S₂. An almost linear relationship between the normalized peak areas (resp. intensities) of the two main absorption features and the PS chain length was found by Pascal *et al.*¹⁵ and this is suggested to be used to resolve different PS from XAS spectrum.

In order to check these theoretical predictions, the measured HERFD XAS spectra were fit to the model consisting of both main absorption features followed by an edge. A single Gaussian was used for the pre-edge absorption peak. Due to the high experimental energy resolution, the main absorption peak reveals a slight asymmetry on the high energy side, which can be accounted for by adding two additional arbitrary Gaussians on the high energy tail of the main Gaussian peak. A cumulative Gaussian distribution was used to describe the edge. This model describes well the measured spectra and is used to extract peak area (and intensity) ratios and energy splitting of the two main absorption peaks (Fig. 3 a).

The experimental values can be directly compared to the theoretical predictions of Pascal *et al.*¹⁵ A comparison of the peak area ratios as a function of PS mean chain length is presented in Fig. 3 b. Very good agreement is obtained for long chain PS Li₂S_x (x≥6), confirming the linear slope predicted by the theory. For shorter chain PS, this slope is less pronounced and the difference between experimental and theoretical values increases. For the shortest Li₂S₂ PS sample, the experimental spectrum still exhibits both absorption features, whereas only the pre-edge peak due to sulfur external atoms should be observed, as no internal sulfur atoms exist in this shortest polysulfide. This difference is attributed to the fact that the standard solutions do not consist of a pure Li₂S_x species, but of lithium PS mixtures in equilibrium with a mean composition Li₂S_x, with the possible presence of precipitated species.³ Indeed, it is well known that lithium PS in solution tend to disproportionate and comproportionate in rapid equilibrium. The presence of chemical equilibria in PS solutions, associated with disproportionation reactions,³ is responsible for the formation of multi PS species in one mean Li₂S_x composition.

Despite that, the ratio of internal/terminal sulfur atoms in solution, which is the experimental quantity determined by HERFD-XAS, is still conserved and reflects the mean PS chain length corresponding to the nominal x value of the initial Li_2S_x standard solution. On the other hand, such chemical equilibria have an indirect impact on the shorter chains solutions, due to the formation of solid species and/or the incomplete reaction of Li with S_8 . Indeed, Li_2S (and possibly Li_2S_2) being less soluble than other PS counterparts,³ the disproportionation reactions lead to the formation of insoluble products, which precipitate from the solution.¹⁷ Due to the precipitation of these species, experimental HERFD spectra correspond to $\text{Li}_2\text{S}_{x'}$ solutions, with mean chain length x' larger than the nominal x value of the short chain PS standard solutions. While the absence of solid species due to the high solubility of PS enables the theoretically predicted ratios to be obtained for long PS chains, the precipitation of less soluble short PS chains explains the discrepancy between experimental values and theory, which increases with decreasing the PS chain length.

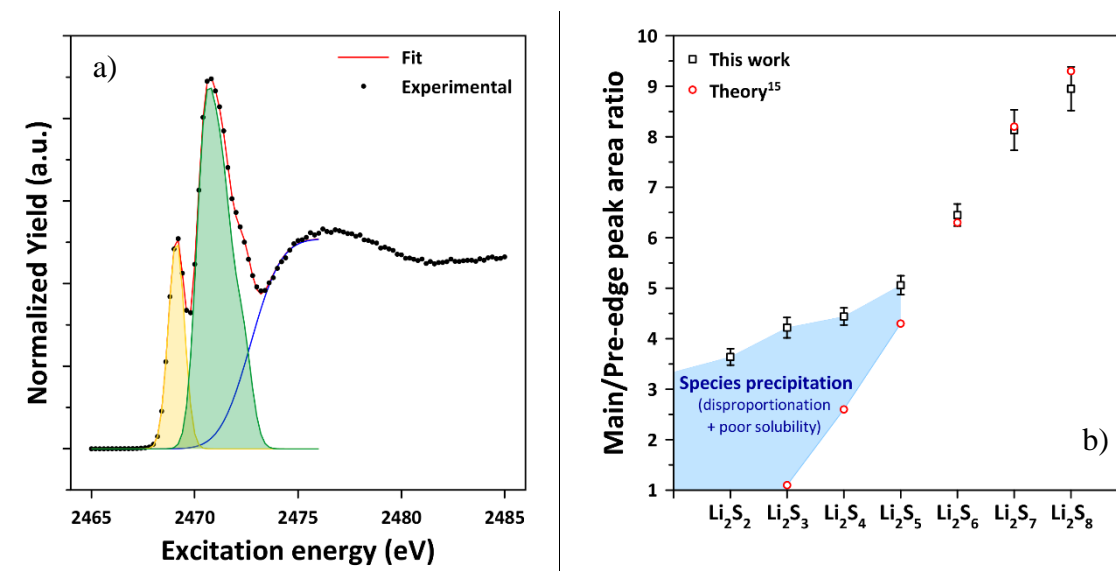


Figure 3. (a) HERFD sulfur K-edge absorption spectrum recorded for the Li_2S_2 standard solution fitted with the model described in the text. (b) Experimental area ratios for the two absorption peaks compared to theoretical values of Pascal *et al.*¹⁵

In addition to peak area ratios, the terminal/internal peak splitting have also been extracted from the spectra yielding an average value of 1.46 ± 0.05 eV for all of the standard PS solutions. This is in agreement with the theoretically predicted 1.5 eV value for the longest Li_2S_8 polysulfide chain.¹⁵ According to theoretical calculations, a correlation between peak splitting and partial atomic charge difference is expected so that the splitting should increase to 2.1 eV approaching Li_2S_3 . However, no significant variation of peak splitting between different measured standards has been observed in the present experiment. Despite significant discrepancies of experimental peak area ratios from the theoretical values observed for short chain PS, at least a slight systematic variation of the peak splitting as a function of the PS chain length would be expected, but this is not supported by our experimental data.

To conclude, the increased spectral resolution of HERFD sulfur K-edge absorption spectra of standard PS samples has allowed the clear separation of the pre-edge absorption peak corresponding to terminal sulfur atoms from the main absorption peak due to internal atoms. This allows the quantitative evaluation of the evolution of the peak area ratio as a function of the PS mean chain length. Results of this experimental work generally confirm the theoretical predictions of Pascal *et al.* that the normalized area of the pre-edge is a reliable fingerprint of Li_2S_x mean chain length. Similar to standard XAS, HERFD XAS can be performed in *operando* mode, which opens the perspective to use such information to determine and quantify the soluble intermediate species formed in the battery, complementary to X-ray diffraction, which is principally used to quantify the solids formed during the electrochemical process. Therefore, a complete quantitative analysis of sulfur compounds (liquid and solid phases) formed during the electrochemical process is now feasible.

EXPERIMENTAL METHODS

Lithium polysulfide (PS) standard solutions were prepared in an argon-filled glovebox by mixing elemental sulfur (dried, 325 mesh, Alfa Aesar) and metallic lithium in a mixture of tetraethylene glycol (TEGDME, dried, 99%, Aldrich) and 1,3-dioxolane (DIOX, anhydrous, 99,8%, Aldrich) with a volume ratio of 1/1. The different lithium PS standard solutions, with equivalent compositions of Li_2S_x ($x = 2, 3, \dots, 8$), were prepared at $10^{-1} \text{ mol.L}^{-1}$, by using stoichiometric amounts of lithium and sulfur powders. The dispersions reacted at 50°C in a closed tube over a few days to ensure stable composition of the solutions.

For the purpose of RIXS measurements, lithium PS standard solutions were enclosed in a CR2032 coin cell designed for XAS experiments. The coin cells were assembled in an argon filled glovebox, the casing was drilled on one side and airtightness was achieved using a Kapton[®] film ($7 \mu\text{m}$), in order to allow X-rays to reach the sample and collect the fluorescence signal.

Measurements were performed at the ID26 beamline of the European Synchrotron Radiation Facility (ESRF) equipped with a double Si(111) monochromator providing an energy resolution of $\sim 0.35 \text{ eV}$ at the sulfur K-edge (2472 eV). The photon beam was focused to $50 \times 250 \mu\text{m}$ and the incoming flux was $\sim 5 \times 10^{12} \text{ photons/s}$. RIXS spectra were recorded by an in-vacuum Johansson type curved-crystal spectrometer for high-energy resolution X-ray emission spectroscopy in the tender range.¹⁸ The spectrometer operates in the dispersive mode using CCD camera containing 770×1152 pixels with $22.5 \mu\text{m}^2$ pixel. The first order reflection of the Si(111) analyzer crystal was used for this experiment. At fixed detector position tuned to central Bragg angle corresponding to the S $\text{K}\alpha$ emission line the spectrometer covered the full bandwidth of $\sim 29 \text{ eV}$ with energy resolution of $\sim 0.4 \text{ eV}$. In addition, a Si photodiode was mounted in the spectrometer chamber to collect also fast XANES spectra, which were used to monitor any possible radiation damage induced by the incident photon beam.

AUTHOR INFORMATION

The authors declare no competing financial interests.

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